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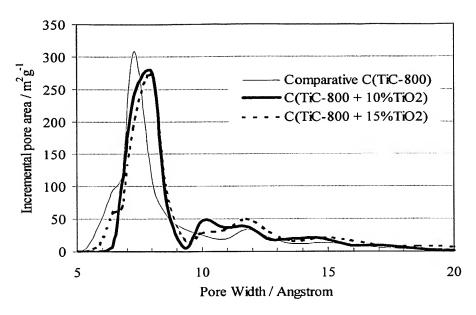
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(54) Title: A METHOD OF MAKING THE POROUS CARBON MATERIAL AND POROUS CARBON MATERIALS PRODUCED BY THE METHOD



(57) Abstract: A method for making the microporous carbon with modified pore size distribution and advanced sorption behaviour. The carbon is derived from metal or metalloid carbides. The method uses of oxidant in reaction medium that during the carbide conversion into carbon expands small micropores, or the temperature T1 in the beginning of the reaction is higher than the temperature T2 in following reaction. Both approaches improve the penetrability in outer layers of the microporous carbon particle and can be used separately of each other or combined. The microporous carbon obtained is free of impurities and possesses extremely narrow pore size distribution.

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# A method of making the porous carbon material and porous carbon materials produced by the method

### FIELD OF INVENTION

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This invention relates to the synthesis of microporous carbon materials with the modified pore size distribution. It also relates to an improvement in the carbon synthesis procedure considering the halogenation of metal or metalloid carbides.

### BACKGROUND OF THE INVENTION

Halogenation of mineral carbides, hereinafter carbides, has been used to produce microporous carbon materials with remarkably narrow pore size distribution as described in US20020097549. According to the international standards (IUPAC) the micropores are with the size of less than 20Å (1Å = 0.1 nm). Pore size maximum of these materials may be as low as 6-7Å. The carbide-derived microporous carbons are very attractive materials in several important application fields like gas and liquid purification from contaminating impurities, specially those having small-size atoms or molecules; storage of small-size molecules such as hydrogen and methane; desalination of drinking water such as method applying flow-throw capacitors; electric energy storing devices such as batteries and capacitors, etc.

The important issue within most of porous carbon applications is the quantity of pores with desired pore size in bulk volume of carbon. In fact the microporous carbide-derived carbon, usually made at temperature below 800-900 °C, contains considerable amount of too small pores below 6-7Å as shown in fig 1, which normally are not accessible in practical adsorption processes. The peak pore size is suggested to be adjusted within few angstroms by changing the carbide chlorination temperature [Gogotsi et al. Nature Mat., Vol. 2, p. 591 (2003)]. However, in practice the increased chlorination temperature reflects not only in the shift of the peak pore size but also in widening of the pore size distribution. The latter fact is usually undesired, while resulting in the lost molecular sieve behaviour of the carbide-derived carbon. Furthermore, the temperature initiated pore size

shift not necessarily avoids formation of the inaccessible very small micropores. One can achieve substantial shift in the average pore size at reaction temperatures above 900-950 °C, however, the high temperature values support the rearrangement of the carbon atoms into higher ordered nanostructures, thereby noticeably decreasing the quantity of micropores and the active surface in total volume of the carbon.

Prior art in PCT/EP03/04202 teaches how to enlarge the small micropores in highly microporous carbide-derived carbon by subsequently performed specific oxidation in very small micropores. However, when the carbon to be treated is too nanoporous and therefore difficult to saturate with the oxidising agent, the method is not effective enough or needs to be repeated several times. This results in the undesired resources consumption and increased cost of the final carbon material.

It is thus an object of this invention to provide a cheaper and more productive method for making the microporous carbon-derived carbon with improved access in micropores.

It is a further object of this invention to provide an improved carbon with a narrow pore size distribution.

# SUMMARY OF THE INVENTION

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The above objectives of the invention are achieved by modifying the long-known method of halogenation of the mineral carbides. More precisely, the process of extracting the element other than carbon from the crystal lattice of mineral carbide is modified so that etching of the carbide with halogen containing gas leaves a microporous carbon material with improved pore size distribution and advanced sorption properties.

This invention provides a method to particularly enlarge small micropores using in situ oxidation during the carbide halogenation process or controlling the temperatures during the carbide halogenation process. Invention regards to the modifying of the carbide halogenation process to improve the microporous structure of carbon produced during the process. General mass balance of carbon formation from carbides is described by the following equation:

$$M_vC + yz/2X_2 \rightarrow C + yMX_z$$
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where subscripts are stoichiometrical constants, X<sub>2</sub> corresponds to a halogen, preferably chlorine, and M denotes the metal or metalloid. The bulk microporous structure is produced from most of the stoichiometric carbides at temperatures below 800-900 °C. Above this temperature increases the tendency of forming the multi-layered nanographitic lamellas and sheets that creates the larger micro- and mesopores in carbon. Below 800-900 °C the microporous carbon with peak pore size of 7-8 Å is formed, which still contains considerable amount of pores less than 6-7 Å. Due to the high order of homogeneity these small micropores are hardly accessible to any sorbing molecules especially from liquid phase. Prior art describes enlarging of these pores with post oxidation of carbon. This invention describes the method that particularly avoids formation of inaccessible micropores during carbide conversion into microporous carbon.

The essence of invention is the oxidative or temperature enlargement of small micropores at surface layers of carbon particles that is in situ performed during the process of carbide conversion into the carbon. In the first embodiment of the invention the predetermined amount of the oxidising reagent is led into the contact with precursor carbide prior to the contact with halogen gas, whereby the oxidising reagent interacts with halogen releasing the oxygen that subsequently oxidises the carbon. Oxidation rate is controlled by the amount of oxidising reagent mixed with precursor carbide. In the second embodiment of the invention the method bases in the fact that the average size of the carbon pores depends on the temperature of the carbide halogenation. The depth and rate of the surface modification thereby is controlled by the thermal conditions of the carbide etching.

In a preferred embodiment the carbide or a mixture of different carbides, characterised by the formula M<sub>y</sub>C, is chosen among those characterised by the stoichiometric constant y < 1.5 and preferably among those characterised by the stoichiometric constant y < 1.1. More specifically, the carbide-forming element M is preferably Titanium (Ti), Silicon (Si) or Aluminium (Al) or it denotes the mixture of different elements, preferably among Ti, Si, and Al. A general feature of carbides suitable for the present invention is that at temperatures below 800 °C, they yield

carbon with the relative content of micropores >95% and with dominating pore size below 9 Å.

The invention also relates to a microporous carbon material having comparatively larger pores in surface layers and narrower pores in the inner layers of the carbon particle; a density of compacted carbon powder of at least 0.6 g/cm<sup>3</sup>, a specific surface area of 1000-2200 m<sup>2</sup>/g and a relative specific surface area by pore size showing a maximum peak within the pore size range 7.5-20 Å according to the low-temperature nitrogen adsorption using density functional theory.

#### **DESCRIPTION OF THE DRAWINGS**

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The present invention will be prescribed in more detail from the following description and examples of a not limiting the embodiment with reference to the attached drawings, wherein

Fig 1 shows the comparison of DFT pore size distributions of the carbide-derived microporous carbons according to an first embodiment of the invention and that known from the prior art;

Fig 2 shows the cumulative pore area plots of carbide-derived microporous carbons according to first embodiment of the invention and that known from the prior art;

Fig 3 shows HRTEM image of microporous carbon according to the first embodiment of the invention;

Fig 4 shows HRTEM image of the material according to the prior art;

Fig 5 shows the comparison of DFT pore size distributions of the carbide-derived microporous carbons according to the second embodiment of the invention and that known from the prior art;

25 Fig 6 shows the cumulative pore area plots of carbide-derived microporous carbons according to the second embodiment of the invention;

Fig 7 shows HRTEM image of the microporous carbon formed at the temperature T2 of the synthesis;

Fig 8 shows HRTEM image of the microporous carbon formed at the temperature T1 of the synthesis; and

Fig 9 shows the HRTEM image of the microporous carbon formed during step-bystep halogenation of the carbide at the temperatures T1 and T2.

#### DESCRIPTION OF THE PREFERRED EMBODIMENTS

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The bulk microporous carbon structure is produced from most of the stoichiometric mineral carbides at temperatures below 800-900 °C. General mass balance of carbon formation from carbides is described by the following equation (Eq. 1):

$$M_{v}C + yz/2X_{2} \rightarrow C + yMX_{z} \tag{1}$$

where subscripts are stoichiometrical constants, X<sub>2</sub> corresponds to a halogen, preferably chlorine, and M denotes the metal or metalloid. Below 800-900 °C the microporous carbon with peak pore size of 7-8 Å is formed, which still contains considerable amount of pores less than 6-7 Å. Due to the high order of homogeneity considerable quantity of micropores are useless in such carbon materials. These small micropores are hardly accessible to any sorbing molecules especially from liquid phase. This invention provides a method to particularly enlarge small micropores using in situ oxidation during the carbide halogenation process or controlling the temperatures during the carbide halogenation process. Invention regards to the modifying of the carbide halogenation process to improve the microporous structure of carbon produced during the process. This invention describes the method that particularly avoids formation of inaccessible micropores during carbide conversion into microporous carbon.

The novelty of method is that during halogenation of the mineral carbide particles the carbon particles are formed, which have modified microporous structure. The essence of the modified pore structure is that the density of the outer layers of carbon particles is somewhat lower than the density in inner layers of carbon particles, with that improving the penetrability and access of micropores by sorbing

substance (gaseous or liquids). The lower density of outer layers compared to the inner layers of carbon particle is achieved by applying the limited amount of the oxidizing reagent during carbon formation from the carbide or by applying for short predetermined period the reaction temperature T1 in initial stage of the carbide halogenation, whereby T1 is somewhat higher than the temperature T2 applied in subsequent halogenation of the carbide until desired rate of the carbide conversion into carbon is completed. It is possible to use the oxidizing agent and temperatures T1 and T2 in the same carbide halogenation process.

The carbon having modified micropore structure according to this invention is more effective adsorbent for the gas-storage, purification and specific extraction processes. These carbon materials are also more suitable than the carbon known from prior art for use in the energy storing devices. The improved characteristics regarding to the application in the energy storing devices include:

- an increase of the specific capacitance of the carbon,

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- an increase of the deliverable energy density of the carbon.

In a preferred embodiment the carbide or a mixture of different carbides, characterised by the formula  $M_yC$ , is chosen among those characterised by the stoichiometric constant y < 1.5 and preferably among those characterised by the stoichiometric constant y < 1.1. More specifically, the carbide-forming element M is preferably Titanium (Ti), Silicon (Si) or Aluminium (Al) or it denotes the mixture of different elements, preferably among Ti, Si, and Al. A general feature of carbides suitable for the present invention is that at temperatures below 800 °C, they yield carbon with the relative content of micropores >95% and with dominating pore size below 9 Å.

25 For example the mass balance of titanium carbide chlorination reaction is expressed by the following equation (Eq. 2):

$$TiC + 2Cl_2 \rightarrow C + TiCl_4 \tag{2}$$

The equation assumes that theoretically one mole of carbon could be produced from one mole of titanium carbide. In reality, the part of carbon is removed as carbon tetrachloride in secondary reaction, due to the excess of chlorine in reaction medium. The rate of the secondary reaction considerably depends on the temperature in reaction medium. In examples of the present invention approximately 14% loss of carbon is considered in chlorination of pure TiC at 800 °C. In addition, the yield of the carbon depends between the 70-90% from the type of reactor, type of the raw material and from the terms of synthesis.

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In prior art it is known that the carbon derived from TiC at 800 °C is almost completely amorphous microporous according X-ray diffraction and the high-resolution transmission electron microscopy studies. Characteristic HRTEM image is shown in Fig 4. Such carbon possesses high BET surface area of approximately 1350 m² per gram and is almost completely microporous with a peak pore size of ~7 Å (Fig 1). Up to 40% of surface area in this carbon material relates to the micropores, which size is less than 7 Å (Fig 2). These very small pores are rather often useless in practical adsorption processes while being inaccessible to adsorbate molecules.

In the other hand, it is well known the method, also used as a part of so called Kroll process for Titanium production, which consists in passing chlorine over ilmenite or rutile, heated to redness with carbon. Respective overall reaction could be described by the following equation (Eq. 3):

$$C + TiO2 + 2CI2 \rightarrow TiCI4 + CO2$$
 (3)

According to this equation one mole of TiO<sub>2</sub> oxidizes one mole of carbon. Although the intermediate products in this process could be TiC, CO and O<sub>2</sub>, finally the volatile TiCl<sub>4</sub> and CO<sub>2</sub> are formed in chlorine flow.

To minimize the formation of undesirably small micropores in carbon during carbide chlorination the present invention applies to combining the chlorination of carbide (Eq. 2) and in situ oxidation of carbon (Eq. 3) that is performed by mixing of oxide with precursor carbide and subsequently halogenizing this mixture. The following example describes the chlorination of the mixture of TiC and TiO<sub>2</sub> in stationary bed reactor at 800 °C.

In the other hand halogenation of TiC at the temperature 950-1000 °C there will be also formed prevailingly the microporous carbon. It is also known that with

increasing the temperature increases tendency of forming of the ordered nanostructures such as graphitic sheets and lamellas and therefore the formed carbon has so-called transportation channels and the medium size of the pores shifts to the larger values. Characteristic HRTEM image is shown in Fig 8.

To assure the admission to the small micropores the second embodiment of the present invention uses the step-by-step halogenation of the carbide. The halogenation will start at the temperature T1 according to the Eq. 2, wherein T1 is higher than it is necessary for forming a homogenous structure of the microporous carbon, and thereafter the halogenation will be continued at the temperature T2 to form a microporous carbon (for example chlorination of the TiC not over 800 °C).

Although the following examples are based in chlorination of the Titanium carbide for obtaining the microporous carbon the basic carbide can be any metal or metalloid carbide for example silicon carbide, aluminium carbide, molybdenum carbide or boron carbide. In a preferred embodiment the carbide or a mixture of different carbides, characterised by the formula  $M_yC$ , is chosen among those characterised by the stoichiometric constant y < 1.5 and preferably among those characterised by the stoichiometric constant y < 1.1. More specifically, the carbide-forming element M is preferably Titanium (Ti), Silicon (Si) or Aluminium (Al) or it denotes the mixture of different elements, preferably among Ti, Si, and Al.

# 20 EXAMPLE 1

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According to the first embodiment of the invention the titanium carbide (H.C. Starck, grade C.A., 50 g) with an average particle size of 1.3-3 microns was thoroughly mixed with Titanium oxide powder (Kemira, 2.5g). This mixture was thereafter loaded into a quartz stationary bed reactor and let to react with a flow of chlorine gas (99.999% assay) for 3h at 800 °C. Flow rate of chlorine gas was 1.5 l/min. The by-product, TiCl<sub>4</sub>, was led away by the stream of the excess chlorine and passed through a water-cooled condenser into a collector. After that the reactor was flushed with Argon (0.5 l/min) at 1000 °C for 1h to remove the excess of chlorine and residues of gaseous by-products from carbon. During heating and cooling, the reactor was flushed with a slow stream (0.5 l/min) of argon. Resulting carbon powder (9.6g) was moved into quartz stationary bed reactor and treated

with hydrogen gas at 800 °C for 3.5h to dechlorinate deeply the carbon material and also to remove the oxygen containing functional groups from the surface of carbon. During heating and cooling, the reactor was flushed with a slow stream of Argon (0.3 l/min). Final yield of the carbon material was 8.5g (85% of theoretical).

The rate of oxidation is controlled by the amount of oxide added to the precursor carbide. The overall yield of carbon is in good agreement with the mass balance in Eq. 2 as shown with the data in Table 1.

TABLE 1. The yield of the carbon

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Entry	TiO <sub>2</sub>	Theoretical yield of C	Experimental yield of C	
	[%wt. per TiC]	[%]*	[%]	
1	5	82	85	
2	10	79	80	
3	15	75	75	

<sup>\*</sup> Theoretical yield is calculated considering the fact that the average yield of carbon derived from 1 mole of TiC with the absence of TiO<sub>2</sub> is 0.86 mole.

Porous structure of carbon materials produced by varying the precursor carbide, oxide content and chlorination temperature were characterized using different nitrogen and benzene sorption methods. Low temperature nitrogen sorption experiments were performed using Gemini Sorptometer 2375 (Micromeritics). The specific surface area of carbon materials was calculated according BET theory up to the nitrogen relative pressure (P/P<sub>0</sub>) of 0.2. The total volume of pores was calculated from nitrogen adsorption at relative pressure (P/P<sub>0</sub>) of 0.95. Pore size distribution was calculated from low-temperature nitrogen adsorption isotherm according to the Barrett-Joyner-Halenda (BJH) method and the method based on Density Functional Theory (DFT).

Adsorption dynamics of benzene vapours was studied at room temperature using computer controlled weighing of the carbon samples in benzene vapours at normal pressure and room temperature in time. The volume of pores that adsorbed

benzene under the above-described conditions was calculated according to the equation:

$$W_{s} = (m_{2} - m_{1}) / m_{1} \times d_{C_{a}H_{a}}$$
(4)

where  $m_1$  and  $m_2$  are the initial and final weights of the test-sample, respectively, and  $d_{C6H6}$  is the density of benzene at room temperature.

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Specific surface ( $S_{BET}$ ) according to BET, pore volume according to benzene ( $W_s$ ) and nitrogen ( $V_{tot}$ ) and incremental pore volume ( $V_{d<11A}$ ) according to BJH method of carbon of the invention compared to the prior art are presented in Table 2.

TABLE 2. The specific characteristics of the carbon according to the invention

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Carbon	Carbide	Т	TiO <sub>2</sub>	$W_s$	$S_{BET}$	$V_{tot}$	$V_{d \le 11A}$	$V_{d>11\text{\AA}}$
		[°C]	[%]	[cm <sup>3</sup> /g]	[m <sup>2</sup> /g]	[cm <sup>3</sup> /g]	[cm <sup>3</sup> /g]	[cm³/g]
Comparison 1	TiC	800	-	0.59	1445	0.68	0.41	0.27
Invention 1a	TiC	800	5	0.60	1512	0.70	0.41	0.29
Invention 1b	TiC	800	10	0.66	1538	0.74	0.36	0.38
Invention 1c	TiC	800	15	0.74	1695	0.80	0.35	0.45
Comparison 2	TiC	700	_	0.54	1292	0.63	0.40	0.23
Invention 2a	TiC	700	10	0.62	1503	0.70	0.40	0.30
Comparison 3	SiC	800	_	0.50	1209	0.59	0.36	0.23
Invention 3a	SiC	800	10	0.56	1347	0.67	0.36	0.31
Invention 3b	SiC	800	15	0.57	1389	0.69	0.35	0.34
Comparison 4	SiC	900	-	0.49	1218	0.60	0.38	0.23
Invention 4a	SiC	900	10	0.50	1283	0.64	0.36	0.27

The pore size distribution presented in Fig 1 and Fig 2 according to the Density Functional theory confirms the modified microporous structure of carbon of this invention compared to that of the prior art. The effect of oxidation during chlorination is also obvious in HRTEM picture in Fig 3. The bulk carbon of this invention is notably coarser compared to the material of prior art (Fig 4).

## EXAMPLE 2

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According to the second embodiment of the invention the fluidized bed reactor was loaded with titanium carbide (Pacific Particulate Materials, 400g) with an average particle size of ~70 µm and flowed with the Argon until the reaction temperature was achieved. Thereafter the carbide was let to react with a flow of chlorine gas (99.999% assay) for 15 min at 950 °C. Flow rate of chlorine gas was 7 l/min. The by-product, TiCl4, was led away by the stream of the excess chlorine and passed through a water-cooled condenser into a collector. After that the reactor was flushed with Argon (6 I/min) for 20 min and the temperature of reactor thereafter was decreased to 800 °C. At 800 °C, chlorination of the carbide was completed using chlorine flow of 7 l/min. Thereafter again the reactor was flowed with Argon (3 l/min) and flushed at 1000 °C 60 min. Resulting carbon powder (61 g) was moved into quartz stationary bed reactor and treated with hydrogen gas at 800 °C for 4h to dechlorinate deeply the carbon material and also to remove the undesired functional groups from the surface of carbon. During heating and cooling, the reactor was flushed with a slow stream of Argon (0.3 I/min). Final yield of the carbon material (invention 7a) was 59g (73.5% of theoretical).

The step-by-step chlorination procedure of TiC was also carried out in the rotary kiln reactor using the TiC powder with the fine particles (H. C. Starck, type CA,  $1.3-3 \mu m$ ) and the carbon materials (invention 5a and 6a) were compared with the carbon materials know in the prior art synthesised at discrete temperatures.

Specific surface ( $S_{BET}$ ) according to BET, pore volume according to benzene ( $W_s$ ) and nitrogen ( $V_{tot}$ ) sorption and incremental pore volume ( $V_{d<11A}$ ) according to BJH method of carbon of the invention compared to the prior art are presented in Tables 3 and 4.

Table 3. Carbon material characteristics when the method is carried out in the fluidized bed reactor

Carbon	T1	T2	Ws	S <sub>BET</sub>	$V_{tot}$	V <sub>d&lt;11</sub> A	V <sub>d&gt;11A</sub>
	°C	°C	cm³/g	m²/g	cm³/g	cm³/g	cm³/g
Comparison 5	-	700	0.54	1292	0.63	0.40	0.23

Invention 5a	950	700	0.66	1396	0.78	0.37	0.41
Comparison 6	-	800	0.60	1348	0.67	0.37	0.30
Invention 6a	950	800	0.65	1474	0.74	0.39	0.35
Comparison 6b	950	-	0.67	1572	0.77	0.33	0.44

Table 4. Carbon material characteristics when the method is carried out in the rotary kiln reactor

Carbon	T1	T2	Ws	S <sub>BET</sub>	$V_{tot}$	V <sub>d&lt;11Å</sub>	V <sub>d&gt;11Å</sub>
	°C	°C	cm <sup>3</sup> /g	m²/g	cm³/g	cm³/g	cm³/g
Comparison 7	-	800	0.60	1473	0.67	0.36	0.31
Invention 7a	950	800	0.69	1582	0.73	0.34	0.39
Comparison 7b	950	-	0.61	1445	0.67	0.30	0.37

The pore size distribution presented in Fig 5 according to the Density Functional theory confirms the modified microporous structure of carbon of this invention compared to that of the prior art. The effect of step-by-step chlorination method using reaction temperatures T1 and T2 is also obvious in HRTEM picture in Fig 9, which shows the areas of different density in carbon particle as a result of T1 and T2. Material of this invention is clearly different from that known from prior art (cf. Fig. 8).

#### Claims

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- 1. A method of making the porous carbon material with modified content of micropores comprising the step of selecting the temperature T of the halogenation of substrate, which contains the metal or metalloid carbides, wherein metal or metalloid carbides are converted into carbon with the dominating pore size of 7 Å and over.
- 2. A method according to the claim 1 characterized by
- a) making the homogeneous mixture of the metal or metalloid carbide with the oxidizing reagent;
- b) reacting the mixture with the halogen, preferably with the chlorine gas, at temperature T required to convert the metal or metalloid carbide into carbon.
  - 3. A method according to the claim 1 characterized by
  - a) step-by-step halogenation of the substrate containing metal or metalloid carbide, whereas the temperature T of the reaction medium is modified during the synthesis, so that the temperature T1 in the initial phase of the halogenation is higher than required for forming the micropores;
  - b) the further process of the halogenation is implemented thereafter in the temperature T2, which is lower than temperature T1, but sufficient to convert the carbide into microporous carbon.
- 4. The method according to claim 1, wherein the metal or metalloid carbide is selected such that is capable in reaction with the halogen to convert into microporous carbon, whereas the metal or metalloid carbide is preferably selected from the group of TiC, SiC, Al<sub>4</sub>C<sub>3</sub>, Mo<sub>2</sub>C and B<sub>4</sub>C.
  - 5. The method according to claim 4, wherein the metal or metalloid carbide is selected from the group of TiC, SiC, Al<sub>4</sub>C<sub>3</sub>, but most preferably being TiC.
    - 6. The method according to claim 1 or 2, wherein the temperature T of halogenation, which is used for producing the microporous carbon from the metal or metalloid carbide is selected from the range 700 °C to 900 °C.

- 7. The method according to claim 6, wherein the temperature T of halogenation is preferably up to 800 °C.
- 8. The method according to the claim 2, wherein the oxidizing reagent is metal oxide.
- 5 9. The method according to the claim 8, wherein the oxidizing reagent is TiO<sub>2</sub>.
  - 10. The method according to the claim 9, wherein the amount of the  $TiO_2$  in the homogeneous mixture is less than 25%, but most preferably between 5 to 20% in weight relative to the carbide.
- 11. The method according to claim 3, wherein the temperature T1 of halogenation
   in the initial phase of the halogenation is higher than 900 °C, whereas preferably the temperature T1 of halogenation is selected from the range 900 °C to 1100 °C.
  - 12. The method according to claim 11, wherein the temperature T1 of halogenation of the TiC is preferably 950 °C to 1000 °C.
- 13. The method according to claim 3, wherein the temperature T2 of halogenation implemented thereafter in the halogenation is lower than 900 °C, whereas preferably the temperature T2 of halogenation is selected from the range 500 °C to 900 °C.
  - 14. The method according to claim 13, wherein the temperature T2 of halogenation of the TiC is preferably up to 800 °C.
- 20 15. The method according to the any of the preceding claims comprising the steps of

- a) making the homogeneous mixture of the metal or metalloid carbide with the oxidizing reagent;
- b) reacting the mixture with halogen, preferably with the chlorine gas, at temperature T required to convert the metal or metalloid carbide into carbon;
- c) step-by-step halogenation of the metal or metalloid carbide, whereas the temperature T of the reaction medium is modified during the halogenation, so that the temperature T1 in the initial phase of the halogenation is higher than required for forming the micropores,

d) the further process of the halogenation is implemented thereafter in the temperature T2, which is lower than temperature T1, but sufficient to convert the carbide into microporous carbon.

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- 16. The carbon material according to the method prescribed in the claims 1, 2, 4-10 which specific surface area contains at least in extent of 80% micropores with pore size more than 7 Å.
  - 17. The carbon material according to the claim 16, wherein the specific surface area contains at least in extent of 85% micropores with pore size more than 7 Å, from which more than 60% is with the pore size 7-10 Å.
- 18. The carbon material according to the method prescribed in the any of the claims 1, 3, 4, 5, 11, 12 which specific surface area is at least 100 m²/g greater than specific surface area of the carbon material produced by halogenation of the same primary substance only in the temperature T, whereas the specific surface area of the carbon material contains at least in extent of 90% the micropores.

## 16 AMENDED CLAIMS

# [Received by the International Bureau on 10 October 2005 (10.10.2005): original claims 1-18 replaced by amended claims 1-17 (3 pages)]

- 1. A method of making the microporous carbon material with the predetermined arrangement of pore-sizes in carbon particles comprising the step-by-step selection of the temperature T of the halogenation for the heat-treatment in the halogen rich atmosphere of a substrate, which contains the metal or metalloid carbides, wherein metal or metalloid carbides are converted into carbon with the dominating pore size of 7-8 Å inside the carbon particles and over 8 Å in surface layers of the carbon particles.
- 2. A method according to the claim 1 characterized by

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- a) making the homogeneous mixture of the metal or metalloid carbide with the solid oxidizing reagent, which decomposes to the volatile components during the carbide conversion into the carbon;
  - b) reacting this homogeneous mixture with the halogen, preferably with the chlorine gas, at temperature T required to convert the metal or metalloid carbide into carbon.
  - 3. A method according to the claim 1 characterized by that
  - a) during the halogenation of the substrate containing metal or metalloid carbide the temperature T of the reaction medium is modified so that the temperature T1 in the initial phase of the halogenation is higher than required for forming the micropores;
  - b) the further process of the halogenation is implemented thereafter in the temperature T2, which is lower than temperature T1, but sufficient to convert the carbide into microporous carbon.
  - 4. The method according to claim 1, wherein the metal or metalloid carbide is selected such that is capable in reaction with the halogen to convert into microporous carbon, whereas the metal or metalloid carbide is preferably selected from the group of TiC, SiC, Al<sub>4</sub>C<sub>3</sub>, Mo<sub>2</sub>C and B<sub>4</sub>C.
  - 5. The method according to claim 4, wherein the metal or metalloid carbide is selected from the group of TiC, SiC, Al<sub>4</sub>C<sub>3</sub>, but most preferably being TiC.

- 6. The method according to claim 1 or 2, wherein the temperature T of halogenation, which is used for producing the microporous carbon with a dominating pore size of 7-8 Å from the metal or metalloid carbide is selected from the range 700 °C to 1100 °C.
- 5 7. The method according to claim 6, wherein the temperature T of halogenation is preferably up to 800 °C.
  - 8. The method according to the claim 2, wherein the oxidizing reagent is metal oxide.
  - 9. The method according to the claim 8, wherein the oxidizing reagent is TiO<sub>2</sub>.
- 10. The method according to the claim 9, wherein the amount of the TiO<sub>2</sub> in the homogeneous mixture is less than 25%, but most preferably between 5 to 20% in weight relative to the carbide.

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- 11. The method according to claim 3, wherein the temperature T1 of halogenation in the initial phase of the halogenation is higher than 900 °C, whereas preferably the temperature T1 of halogenation is selected from the range 900 °C to 1100 °C.
- 12. The method according to claim 11, wherein the temperature T1 of halogenation of the TiC is preferably 950 °C to 1000 °C.
- 13. The method according to claim 3, wherein the temperature T2 of halogenation implemented thereafter in the halogenation is lower than 900 °C, whereas preferably the temperature T2 of halogenation is selected from the range 500 °C to 900 °C.
- 14. The method according to claim 13, wherein the temperature T2 of halogenation of the TiC is preferably up to 800 °C.
- 15. The method according to the any of the preceding claims comprising the steps of
- a) making the homogeneous mixture of the metal or metalloid carbide with the oxidizing reagent;

- b) reacting the mixture with halogen, preferably with the chlorine gas, at temperature T required to convert the metal or metalloid carbide into carbon;
- c) step-by-step halogenation of the metal or metalloid carbide, whereas the temperature T of the reaction medium is modified during the halogenation, so that the temperature T1 in the initial phase of the halogenation is higher than required for forming the micropores,

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- d) the further process of the halogenation is implemented thereafter in the temperature T2, which is lower than temperature T1, but sufficient to convert the carbide into microporous carbon.
- 16. The carbon material according to the method prescribed in the claims 1, 2, 4-10 which specific surface area contains at least in extent of 85% micropores with pore size 7-20 Å, from which more than 60% is with the pore size 7-10 Å.
  - 17. The carbon material according to the method prescribed in the any of the claims 1, 3, 4, 5, 11, 12 which specific surface area is at least 100 m²/g greater than specific surface area of the carbon material produced by halogenation of the same primary substance only in the temperature T2, whereas the specific surface area of the carbon material contains at least in extent of 90% the micropores with pore size below 20 Å, from which more than 70% is with the pore size of less than 10 Å.

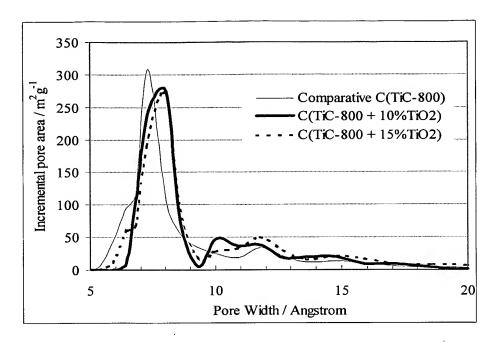


FIG 1

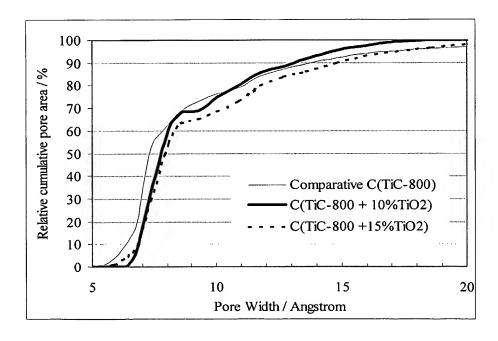


FIG 2

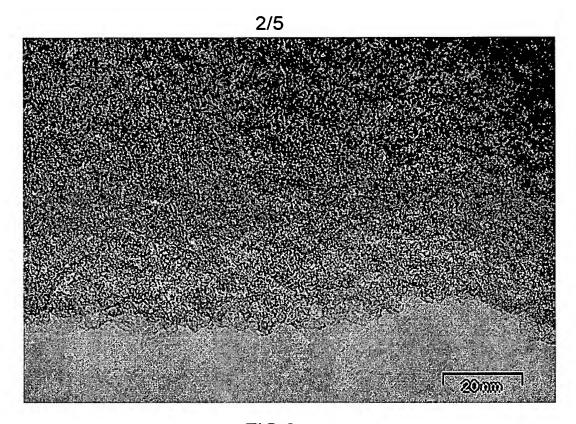


FIG 3

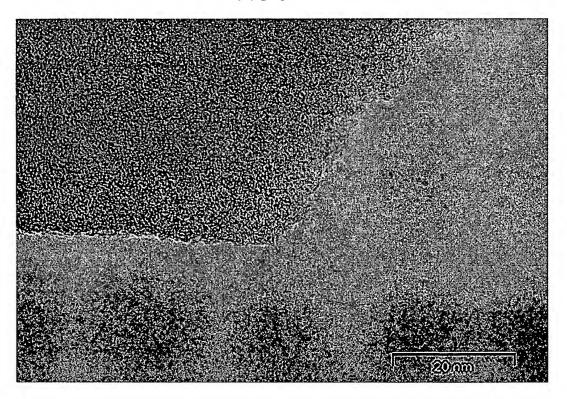


FIG 4

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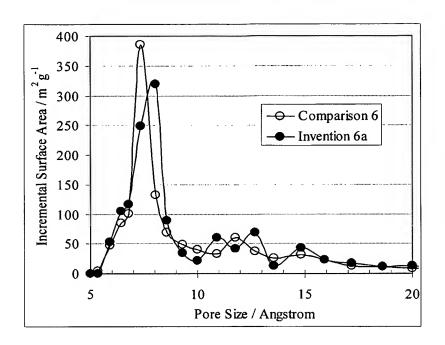


FIG 5

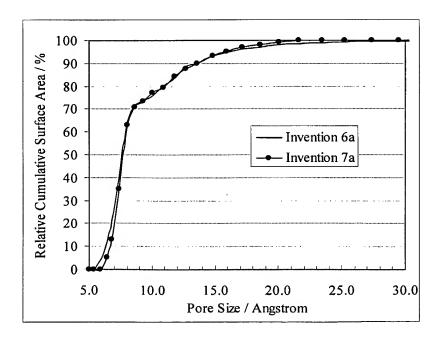


FIG 6

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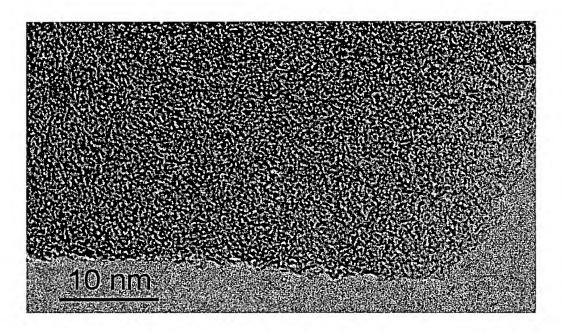


FIG 7

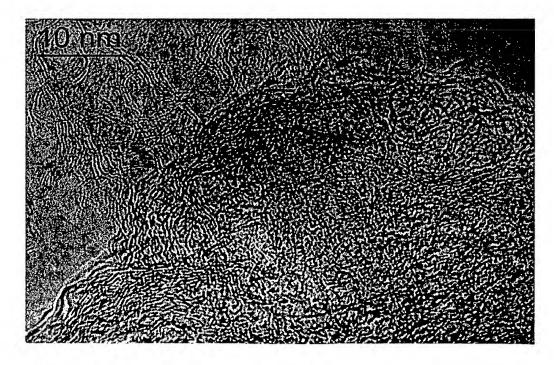


FIG 8

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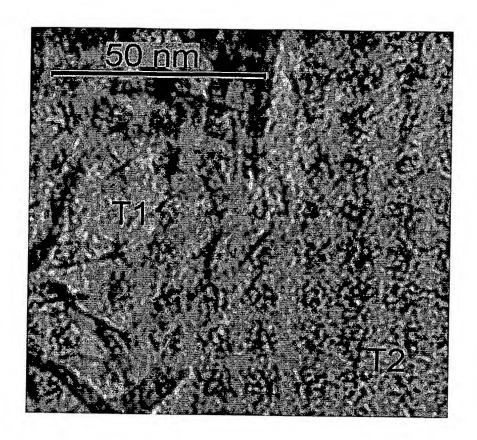


FIG 9

#### INTERNATIONAL SEARCH REPORT

A. CLASSIFICATION OF SUBJECT MATTER IPC 7 C01B31/02 C01E C01B31/10 C01B31/12 According to International Patent Classification (IPC) or to both national classification and IPC **B. FIELDS SEARCHED** Minimum documentation searched (classification system followed by classification symbols) IPC 7 CO1B Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched Electronic data base consulted during the international search (name of data base and, where practical, search terms used) EPO-Internal, WPI Data, INSPEC, COMPENDEX, PAJ C. DOCUMENTS CONSIDERED TO BE RELEVANT Category ° Citation of document, with indication, where appropriate, of the relevant passages Relevant to claim No. X US 2002/097549 A1 (MALETIN YURII ET AL) 1,2,4-8,25 July 2002 (2002-07-25) 16 - 18paragraphs '0033! - '0043!; figure 5; tables 1.3 X LEIS J ET AL: "Carbon nanostructures 1,4,6,7, produced by chlorinating aluminium 16-18 carbide" CARBON. vol. 39, no. 13, November 2001 (2001-11), pages 2043-2048, XP004305010 ISSN: 0008-6223 "Introduction", "Experimental", table 1 X US 5 965 483 A (BAKER ET AL) 16 - 1812 October 1999 (1999-10-12) claims 1,2; table 4 Further documents are listed in the continuation of box C. Patent family members are listed in annex. Special categories of cited documents: "T" later document published after the international filing date or priority date and not in conflict with the application but "A" document defining the general state of the art which is not considered to be of particular relevance cited to understand the principle or theory underlying the invention "E" earlier document but published on or after the international "X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to filing date "L" document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another involve an inventive step when the document is taken alone "Y" document of particular relevance; the claimed invention citation or other special reason (as specified) cannot be considered to involve an inventive step when the document is combined with one or more other such docu-"O" document referring to an oral disclosure, use, exhibition or ments, such combination being obvious to a person skilled in the art. other means "P" document published prior to the international filing date but later than the priority date claimed "&" document member of the same patent family Date of the actual completion of the international search Date of mailing of the international search report 9 August 2005 19/08/2005 Name and mailing address of the ISA Authorized officer European Patent Office, P.B. 5818 Patentlaan 2 NL – 2280 HV Rijswijk Tel. (+31–70) 340–2040, Tx. 31 651 epo nl, Fax: (+31–70) 340–3016 Marucci, A

# INTERNATIONAL SEARCH REPORT

Information on patent family members

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# отчёт о международном поиске

Международная заявка № РСТ/ЕЕ2005/000007

- VIII - COVIA VIII - LVIII -									
А. КЛАССИФИКАЦИЯ ПРЕДМЕТА ИЗОБРЕТЕНИЯ: CO1B31/02 CO1B31/10 CO1B31/12 Согласно международной патентной классификации (МПК-7)									
В. ОБЛАСТИ ПОИСКА:									
Б. ОВЛАСТИ ПОИСКА. Проверенный минимум документации (система классификации и индексы) МПК-7:  CO1В									
Другая проверенная документация в той мере, в какой она включена в поисковые подборки:									
Электронная база данных, использовавшаяся при поиске (название базы и, если, возможно, поисковые термины): EPO-Internal, WPI Data, INSPEC, COMPENDEX, PAJ									
С. ДОКУМЕН	НТЫ, СЧИТАЮЩИЕСЯ РЕЛЕВАНТНЫМИ	1:							
Категория*	Ссылки на документы с указанием, где это частей	возможно, релевантных	Относится к пункту №						
x	US 2002/097549 A1 (MALETIN YURII ET AL) 25 июля 2002 (2002-07-25) абзацы`0033! - `0043!; фигура 5; таблицы 1,3								
х	LEIS J ET AL: "Carbon nanostructu chlorinating aluminium carbide" CAR ноябрь 2001 (2001-11), страницы 20 ISSN: 0008-6223 "Introduction", "Ex	1,4,6,7 16-18							
х	16-18								
графы		данные о патентах-аналог	•						
* Особые категор	рии ссылочных документов:	<ul> <li>Т более поздний документ, опублив приоритета и приведенный для по</li> </ul>	* *						
А документ,	определяющий общий уровень техники		·						
•	ий документ, но опубликованный на дату дной подачи или после нее	X документ, имеющий наиболее бл поиска, порочащий новизну и изс	бретательский уровень						
- ,, ,	У документ, порочащий изобретате с одним или несколькими докуме ванию и т.д.  & документ, относящийся к устному раскрытию, экспониро- ванию и т.д.  & документ, являющийся патентом								
Р документ, о	аналогом								
но после даты испрашиваемого приоритета и т.д. "&" документ, являющийся патентом-аналогом "Р" документ, опубликованный до даты международной подачи, но после даты испрашиваемого приоритета.									
Дата действительного завершения международного поиска: Дата отправки настоящего отчёта о международном поиска: 19.08.2005									
	и адрес Международного поискового органа: атентное Ведомство	Уполномоченное лицо: Магиссі, А							

# ОТЧЁТ О МЕЖДУНАРОДНОМ ПОИСКЕ Информация о патентах-аналогах

Международная заявка № PCT/EE2005/000007

Патентный документ, процитированный в отчёте поиске	Дата публикации	Патент(ы)- аналог(и)	Дата публикации
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